

## The Crystal Structure of Phosphorus Heptabromide, $\text{PBr}_7$

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(Received 11 October 1966)

The crystal structure of phosphorus heptabromide,  $\text{PBr}_7$ , has been determined by single-crystal X-ray diffraction studies. The red compound crystallizes in the orthorhombic space group  $Pnma$  with  $a=9.35 \pm 0.02$ ,  $b=7.94 \pm 0.01$ , and  $c=14.69 \pm 0.02$  Å. The structure contains tetrahedral  $\text{PBr}_4^+$  ions with P-Br distances all about 2.17 Å and an almost linear unsymmetrical  $\text{Br}_3^-$  ion with Br-Br distances of 2.91 and 2.39 Å. The bonding in the tribromide ion is discussed.

### Introduction

Kastle & Beatty (1900) discovered that the supposed red form of  $\text{PBr}_5$  was actually  $\text{PBr}_7$ . Bromine is easily absorbed by  $\text{PBr}_5$  to form the red  $\text{PBr}_7$  and, conversely, is easily lost by  $\text{PBr}_7$  to form  $\text{PBr}_5$ . It seemed likely that the  $\text{PBr}_4^+$  ion of the  $\text{PBr}_5$  (van Driel & MacGillavry, 1943) remained intact in  $\text{PBr}_7$  and that a tribromide ion was formed, in a similar way to the formation of  $\text{PCl}_4^+$  and  $\text{ICl}_2^-$  on addition of  $\text{ICl}$  to  $\text{PCl}_5$  (Zeelezny & Baenziger, 1952). Romers & Keulemans (1958) have reported the structure of a tribromide ion in  $[\text{N}(\text{CH}_3)_3\text{H}^+]_2\text{Br}^-\text{Br}_3^-$ . It was symmetrical and slightly bent ( $171^\circ$ ) with the Br-Br distances both 2.54 Å. An X-ray diffraction study of  $\text{PBr}_7$  was undertaken in order to determine if the tribromide ion existed in  $\text{PBr}_7$  and its configuration if its existence was confirmed.

### Preparation and experimental data

The compound  $\text{PBr}_7$  was prepared in a manner similar to the preparation of  $\text{PBr}_5\text{I}$  (Fialkov & Kuzmenko, 1949) by saturating carbon disulfide with phosphorus pentabromide and slowly adding  $\text{Br}_2$  until crystals appeared. The crystals grew as red flat needles. They were very hygroscopic and lost bromine unless kept in solution or sealed in a glass tube. The crystals can also be prepared by sublimation of a mixture of  $\text{PBr}_5$  and  $\text{Br}_2$  (Kastle & Beatty, 1900). It was found that the 'wet' crystals from solution lasted longer in the air, making it possible to mount one in a Pyrex glass capillary.

The lattice constants for  $\text{PBr}_7$ , determined from Mo  $K\alpha$  precession camera photographs and Cu  $K\alpha$  Weissenberg camera photographs, are  $a=9.35 \pm 0.02$ ,  $b=7.94 \pm 0.01$ , and  $c=14.69 \pm 0.02$  Å. The systematic extinctions ( $k+l=2n+1$  for  $0kl$  reflections,  $h=2n+1$  for  $hk0$  reflections) implied the space group  $Pn2_1a$  or  $Pnma$ .

Owing to the instability of the compound an experimental density was not obtained. However, the only choice for the number of molecules per unit cell that was consistent with the space group and gave a reason-

able value for the calculated density ( $3.60 \text{ g.cm}^{-3}$ ) was 4.

Intensity data were collected on a Weissenberg camera with the multi-film technique using Cu  $K\alpha$  radiation. The size of the crystal used was approximately  $0.6 \text{ mm} \times 0.15 \text{ mm} \times 0.12 \text{ mm}$ . The zero through fifth layers were recorded while rotating around the  $[100]$  direction (needle axis). Of the 761 reflections within the limiting sphere of Cu  $K\alpha$  radiation for the layers recorded, 604 had an intensity greater than the minimum observable intensity. These intensities were measured visually with a set of standard spots made from the same crystal used to collect the intensity data.

### Determination of the structure

It was assumed that the centrosymmetric space group  $Pnma$  was the correct one. This choice was confirmed by the solution of the phase problem, and subsequently by a statistical analysis by Wilson's (1949) method. With eightfold general positions and four molecules per unit cell, it was necessary to place some of the atoms of the  $\text{PBr}_7$  unit in sets of fourfold special positions. Stereochemically, it was necessary to place the  $\text{PBr}_4^+$  moieties on the mirror planes at  $y=\frac{1}{4}$  and  $\frac{3}{4}$  with the phosphorus atom and two of the bromine atoms [labeled Br(2) and Br(3)] located directly on the mirror plane and with the remaining two bromine atoms [labeled Br(1)] related to each other by the mirror plane. For the  $\text{Br}_3^-$  ions, three choices were possible. One choice would locate the central bromine on a center of symmetry, thus requiring the ion to be linear and symmetrical ( $D_{\infty h}$ ). A second choice would locate each tribromide ion across one of the mirror planes with a symmetrical but possibly non-linear ion ( $C_{2v}$ ) resulting. The third choice involved locating all three bromine atoms on a mirror plane. In this case, the ion would not be required to have any symmetry beyond  $C_s$ .

A three-dimensional sharpened Patterson map was calculated with the Fourier summation program of Sly, Shoemaker & Van den Hende (1962) on an IBM 709 computer. The positions of the atoms in the  $\text{PBr}_4^+$  ion were estimated from this map. There were no

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Patterson vectors consistent with the first two possible choices for the location of the tribromide ion. Since no vectors could be found that fit a symmetrical or slightly distorted linear tribromide ion, a Fourier synthesis was computed using only the positions of the atoms in the PBr<sub>4</sub><sup>+</sup> ion. Six extra peaks showed up on the first map and by trial and error it was found which of these were the three bromine atoms in the tribromide ion. The Busing, Martin & Levy (1962) 709-7090 FORTRAN crystallographic least-squares program was used for final refining. The atomic scattering factors used for P and Br were those in Table 3.3.1A in *International Tables for X-ray Crystallography* (1962). The refinement with isotropic temperature factors proceeded to a value of R<sub>1</sub>=0.134 (see Table 1 for definition of R<sub>1</sub>). On refinement of anisotropic temperature factors and utilizing a modified Hughes weighting scheme (Willett, Dwiggens, Kruh & Rundle, 1962), a value of R<sub>1</sub>=0.095 was obtained. Unobserved reflections were assigned an intensity equal to one third of the lowest observable intensity. They were omitted from the refinement if the calculated structure factor was less than the 'observed' value. The values of the scale factors for the different layers of Weissenberg data obtained during the isotropic refinement were not varied during the anisotropic refinement.

After removing 10 reflections which appeared to be suffering from extinction, the value of R<sub>1</sub> reduced to 0.080. Final parameters are listed in Table 1. The tem-

Table 2. *Observed and calculated structure factors*  
The columns contain *k*, 10|F<sub>o</sub>| and 10 F<sub>c</sub>. An unobserved reflection is denoted by a negative sign in front of 10|F<sub>o</sub>|. Reflections suffering from extinction are denoted by \*.

h k l	10 F <sub>o</sub>	10 F <sub>c</sub>	h k l	10 F <sub>o</sub>	10 F <sub>c</sub>	h k l	10 F <sub>o</sub>	10 F <sub>c</sub>
0 1373 1174	5 388 317		5 5 L 3	3 1405 -1714		8 2 622 891		
1 136 -127	6 -108 92		6 775 -700		5 935 892	9 924 -795		
2 382 -567	7 -131 -94		0 1225 1148		6 -89 171	2 1364 -1248		
4 124 326	8 -110 -39		1 90 -77		7 922 -750	4 267 -298		
5 588 -544	9 -484 9		2 1417 1693		8 263 -205	4 244 -531		
7 217 -704	10 950 938		3 458 474		4 950 938	8 217 255		
8 268 178	11 1338 -1394		4 915 -900		1 624 -457	5 704 958		
9 341 -332	12 1500 -442		2 1206 -1411		7 488 407	6 1743 1607		
1 1189 -1709	13 1702 -1552		3 669 585		5 -82 59	3 742 676		
2 2309 -3616	14 861 -817		4 950 938		6 1743 1607	4 267 -298		
3 902 1003	15 400 -815		5 1212 1042		1 1088 1088	8 244 -531		
4 273 -295	16 887 -307		6 189 -447		1 1088 1088	9 244 -531		
5 1006 -1080	17 933 -574		7 1031 1067		2 1302 1238	8 287 -291		
6 1096 -1009	18 973 -574		8 -105 -32		3 1088 1088	9 244 -531		
7 269 326	19 973 -574		0 1874 -2024		4 744 737	5 760 855		
8 356 313	20 1314 1300		1 1743 -1846		5 760 855	6 1743 1607		
9 510 -477	21 1314 1300		2 1743 -1846		6 1743 1607	7 1743 1607		
0 1597 -2576	22 1314 1300		3 1809 2004		7 1743 1607	8 287 -291		
2 2422 4035	23 1314 1300		4 1809 2004		8 287 -291	9 244 -531		
3 2411 2623	24 1314 1300		5 1809 2004		9 244 -531	1 1088 1088		
4 158 -189	25 1314 1300		6 1809 2004		1 1088 1088	2 1302 1238		
5 1244 -1214	26 1314 1300		7 1809 2004		2 1302 1238	3 1088 1088		
6 1528 1665	27 1314 1300		8 1809 2004		3 1088 1088	4 744 737		
7 1258 1876	28 1314 1300		9 1809 2004		4 744 737	5 760 855		
8 760 -713	29 1314 1300		10 1809 2004		5 760 855	6 1743 1607		
9 380 -414	30 1314 1300		11 1809 2004		6 1743 1607	7 1743 1607		
0 1280 -1849	31 1314 1300		12 1809 2004		7 1743 1607	8 287 -291		
1 690 -741	32 1314 1300		13 1809 2004		8 287 -291	9 244 -531		
5 851 -842	33 1314 1300		14 1809 2004		9 244 -531	1 1088 1088		
7 1258 1876	34 1314 1300		15 1809 2004		1 1088 1088	2 1302 1238		
9 415 -530	35 1314 1300		16 1809 2004		2 1302 1238	3 1088 1088		
0 1280 -1849	36 1314 1300		17 1809 2004		3 1088 1088	4 744 737		
1 690 -741	37 1314 1300		18 1809 2004		4 744 737	5 760 855		
5 851 -842	38 1314 1300		19 1809 2004		5 760 855	6 1743 1607		
7 1258 1876	39 1314 1300		20 1809 2004		6 1743 1607	7 1743 1607		
9 415 -530	40 1314 1300		21 1809 2004		7 1743 1607	8 287 -291		

Table 1. *Parameters for PBr<sub>7</sub>*  
Standard deviations are given in parentheses.  
The β<sub>ij</sub> are defined by:  
T = exp ( -β<sub>11</sub>h<sup>2</sup> - β<sub>22</sub>k<sup>2</sup> - β<sub>33</sub>l<sup>2</sup> - 2β<sub>12</sub>hk - 2β<sub>13</sub>hl - 2β<sub>23</sub>kl).

	<i>x</i>	<i>y</i>	<i>z</i>	β <sub>11</sub>	β <sub>22</sub>	β <sub>33</sub>	β <sub>13</sub>	β <sub>12</sub>	β <sub>23</sub>
P	0.1133 (0.0016)	0.25	0.1477 (0.0005)	0.0010 (0.0020)	0.0053 (0.0010)	0.0018 (0.0003)	0.0008 (0.0005)	0.0	0.0
Br(1)	0.2352 (0.0003)	0.0311 (0.0003)	0.1085 (0.0001)	0.0047 (0.0006)	0.0057 (0.0003)	0.0040 (0.0001)	-0.0003 (0.0001)	-0.0007 (0.0003)	-0.0001 (0.0001)
Br(2)	0.0761 (0.0005)	0.25	0.2934 (0.0003)	0.0060 (0.0010)	0.0168 (0.0005)	0.0026 (0.0002)	0.0004 (0.0003)	0.0	0.0
Br(3)	-0.0916 (0.0006)	0.25	0.0773 (0.0003)	0.0043 (0.0010)	0.0164 (0.0006)	0.0038 (0.0002)	-0.0016 (0.0003)	0.0	0.0
Br(4)	0.1818 (0.0005)	0.75	0.3778 (0.0002)	0.0076 (0.0008)	0.0261 (0.0005)	0.0029 (0.0001)	-0.0007 (0.0002)	0.0	0.0
Br(5)	-0.0804 (0.0005)	0.75	0.4848 (0.0002)	0.0039 (0.0009)	0.0075 (0.0005)	0.0031 (0.0001)	-0.0000 (0.0002)	0.0	0.0
Br(6)	0.4031 (0.0006)	0.75	0.2966 (0.0003)	0.0059 (0.0011)	0.0204 (0.0007)	0.0050 (0.0002)	0.0005 (0.0003)	0.0	0.0

$$R_1 = \frac{\sum_{hkl} | |F_{obs}| - |F_{calc}| |}{\sum_{hkl} |F_{obs}|} = 0.080$$

$$R_3 = \frac{\sqrt{\sum_{hkl} w (|F_{obs}| - |F_{calc}|)^2}}{\sqrt{\sum_{hkl} w |F_{obs}|^2}} = 0.102$$

Table 2 (cont.)

2 2293 2293	2 415 455	2 1312 1328	6 186 -183	H# 1 L#13	0 761 -804	H# 1 L#16
3 1090 1071	3 368 -375	1 175 -328	7 -79 -114	0 204 -243	1 204 -243	0 395 575
4 379 -402	4 423 -463	4 109 -385	5 530 439	1 671 516	3 107 516	1 -99 -77
5 640 -600	5 -82 82	6 738 677	0 317 366	2 509 -546	4 326 -518	2 355 355
6 858 966	6 276 300	7 150 -134	1 -89 -152	3 -33 -136	5 220 -194	3 -88 124
7 594 557	7 261 -234	8 174 -344	2 -89 -164	4 385 359	6 201 313	4 318 319
8 390 -433	8 204 -166	H# 4 L#10	3 -108 -54	5 768 759	H# 4 L#14	H# 2 L#16
H# 2 L# 8	H# 3 L# 9	H# 4 L#10	4 302 286	6 325 -293	H# 4 L#14	H# 2 L#16
0 1889 -1826	0 -79 66	1 972 -1010	2 1187 1292	H# 2 L#13	0 1180 -1073	1 423 417
1 -35 261	1 950 846	2 1187 1292	3 982 1071	0 636 713	1 278 -231	2 316 261
2 -12 -17	2 605 -679	4 -109 117	5 475 -466	1 624 689	3 -114 46	3 305 -368
3 344 -318	3 -86 15	6 406 431	0 1103 -1090	2 601 -792	4 214 -219	4 180 -167
4 1047 -1133	4 -90 84	7 659 588	1 109 -329	3 381 -329	5 264 -261	H# 3 L#16
5 -87 -96	5 1010 1183	H# 5 L#10	2 -89 -154	4 595 588	H# 5 L#14	0 -91 -114
6 154 111	6 381 -326	1 606 -929	3 -108 -54	5 713 674	1 -89 -37	1 -89 -37
7 -99 -131	7 359 306	2 1187 1292	4 302 286	6 504 -487	2 102 1091	3 -77 -57
8 213 -202	8 122 202	H# 1 L#12	0 278 -825	0 -97 -110	1 1022 1091	4 100 -100
H# 3 L# 8	H# 4 L# 9	H# 1 L#12	1 -92 45	2 102 1091	3 908 -916	H# 4 L#16
0 3182 -3227	0 463 423	0 278 -825	2 -89 96	4 152 -79	5 453 247	0 -100 76
1 -35 -115	1 255 -176	1 -92 45	3 469 -480	6 264 -293	H# 0 L#16	1 -98 -111
2 1279 1134	2 515 -474	2 789 -194	4 303 278	3 125 -34	1 109 13	2 113 147
3 -83 -82	3 266 281	3 -94 -75	5 669 -550	4 386 362	2 -106 13	H# 0 L#17
4 2036 -2160	4 332 375	4 330 332	6 89 -58	5 364 -256	3 350 263	1 344 337
5 211 -198	5 132 -33	5 -108 -52	7 -65 80	6 191 -205	4 495 415	H# 1 L#17
6 959 845	6 337 321	6 -96 -238	H# 0 L#11	7 402 367	5 247 207	H# 1 L#17
7 -99 -140	7 207 179	7 -65 80	H# 2 L#12	8 -84 -72	6 191 -205	H# 1 L#17
8 659 -663	H# 5 L# 9	H# 2 L#12	1 1346 1266	0 550 495	H# 1 L#15	0 566 546
H# 4 L# 8	0 753 713	0 550 495	1 1478 -1759	1 245 -212	H# 1 L#15	1 282 282
0 1166 1127	1 431 361	1 -92 45	2 1275 -1243	2 292 -254	1 -109 13	2 113 147
1 412 -800	2 -87 186	2 789 -194	3 347 -385	3 125 -34	2 -106 13	H# 0 L#17
2 455 433	3 846 -925	H# 1 L#11	4 303 278	4 386 362	3 350 263	1 344 337
3 856 966	4 -90 84	6 89 -58	5 669 -550	5 364 -256	4 495 415	H# 1 L#17
4 590 640	5 105 -219	0 153 140	6 191 -205	6 191 -205	5 247 207	H# 1 L#17
5 389 -441	6 -99 -57	1 681 610	H# 3 L#12	H# 5 L#13	0 598 567	0 -87 -95
6 128 164	7 852 -777	2 -90 -66	0 495 519	0 327 -345	2 106 76	2 -81 138
7 000 078	H# 5 L# 8	3 496 54	1 881 610	1 -104 83	3 -72 -79	3 -72 -79
H# 5 L# 8	5 269 -231	5 309 240	2 812 926	2 -101 -142	4 386 362	5 -68 -117
0 318 277	0 2312 2526	6 -104 16	3 -91 150	3 -96 240	5 361 302	0 490 420
1 957 1031	2 605 -638	7 452 -535	4 299 248	4 234 -236	6 191 -205	1 -77 -72
2 980 1151	3 1037 1611	H# 2 L#11	5 -100 102	5 361 302	7 -62 163	H# 2 L#17
3 573 -667	4 377 -382	0 404 -400	6 -96 -72	H# 0 L#14	H# 3 L#15	0 490 420
4 -89 106	5 104 106	1 184 180	H# 4 L#12	0 243 309	0 -101 186	2 280 -289
5 1049 873	H# 1 L#10	2 488 478	0 -109 -161	2 1683 1514	1 272 251	3 193 -199
6 405 430	3 909 -421	3 488 478	4 218 -227	3 364 313	2 -89 10	H# 3 L#17
7 268 -220	4 -83 57	4 218 -227	5 644 575	4 386 362	5 247 207	0 367 346
H# 0 L# 9	5 978 1013	5 978 1013	6 301 -259	6 656 -667	4 -80 116	1 -74 -18
1 1519 1437	6 301 -259	6 301 -259	H# 3 L#11	H# 1 L#14	H# 4 L#15	2 437 -647
2 1191 -1088	7 -96 73	0 344 -348	0 344 -348	1 719 -788	0 -115 -184	H# 4 L#17
3 978 1013	H# 1 L#10	1 480 542	1 480 542	2 820 800	1 260 305	0 483 412
4 746 -474	H# 2 L#10	2 174 189	2 174 189	3 660 684	2 -109 -90	1 77 36
H# 1 L# 9	3 530 -255	3 530 -255	4 238 -241	4 735 711	3 100 -45	2 369 -567
0 1194 -1032	4 238 -241	4 238 -241	5 592 547	5 592 547	4 -86 -108	H# 0 L#18
1 551 -533	5 592 547	6 -95 129	6 -95 129	7 -76 -95	H# 5 L#15	H# 0 L#18
2 1210 1183	6 -95 129	7 -76 -95	7 -76 -95	8 -84 -72	0 343 -371	0 -70 44
3 456 -421	7 -76 -95	8 -84 -72	H# 2 L#14	0 1116 1164	1 398 358	2 270 549
4 754 -806	8 -84 -72	9 -84 -72	0 1116 1164	1 -68 -45	2 -81 104	H# 1 L#18
5 853 -955	9 -84 -72	0 598 621	1 -68 -45	2 -105 95	3 -72 -104	0 167 147
6 689 662	0 598 621	1 875 -917	2 -105 95	3 555 489	4 735 711	1 026 -607
7 629 -640	1 875 -917	2 -109 -19	4 735 711	5 249 -249	H# 0 L#16	0 -99 76
8 363 -427	2 -109 -19	3 234 -272	5 249 -249	6 205 -145	H# 2 L#18	H# 2 L#18
H# 2 L# 9	3 234 -272	4 344 482	6 205 -145	7 -77 95	0 -328 -416	0 -328 -416
0 783 -885	4 344 482	5 1015 -995	H# 3 L#14	H# 3 L#14		
1 349 291	5 1015 -995					

Table 3. Bond lengths and angles in PBr<sub>7</sub>

Bond lengths	
P-Br(1)	2.16 ± 0.01 Å
P-Br(2)	2.17 ± 0.01
P-Br(3)	2.18 ± 0.01
Br(4)-Br(5)	2.91 ± 0.01
Br(3)-Br(6)	2.39 ± 0.01

Bond angles	
Br(1)-P-Br(2)	110.4 ± 0.3°
Br(1)-P-Br(3)	109.8 ± 0.3
Br(2)-P-Br(3)	109.1 ± 0.5
Br(1)-P-Br(1)	107.3 ± 0.5
Br(3)-Br(4)-Br(6)	177.3 ± 0.2

perature factors cannot be considered physically meaningful since no absorption corrections were made on the intensity data. The observed and calculated structure factors are listed in Table 2.

Bond lengths and angles with standard deviations are listed in Table 3. One half of the unit cell is shown in Fig. 1.

Discussion and conclusion

Phosphorus heptabromide exists in the crystalline state as discrete PBr<sub>4</sub><sup>+</sup> and Br<sub>3</sub><sup>-</sup> ions. The overall structure is essentially the same as in NH<sub>4</sub>I<sub>3</sub> (Mooney, 1935), CsI<sub>3</sub> (Tasman & Boswijk, 1955), and CsI<sub>2</sub>Br (Carpenter, 1966). All of the compounds have the same space group

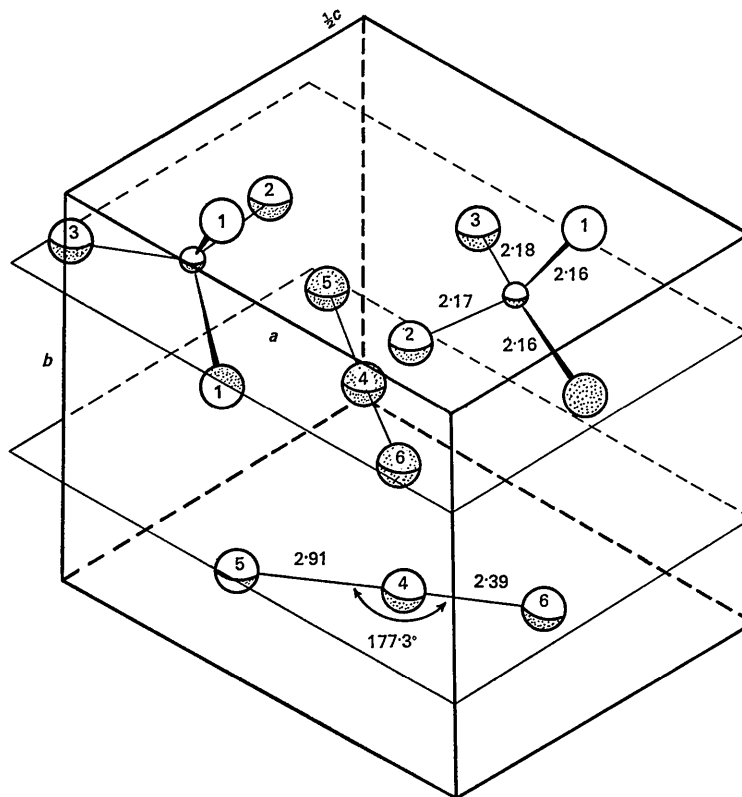


Fig. 1. Illustration of the structure of PBr<sub>7</sub>.

if the axes are chosen similarly. The larger cations in  $\text{PBr}_7$  expand the unit cell mainly in the  $c$  direction (for  $Pnma$ ).

The  $\text{PBr}_4^+$  ion in  $\text{PBr}_7$  is very similar in configuration to the  $\text{PBr}_4^+$  ion in  $\text{PBr}_5$ . It is nearly tetrahedral with all angles in the range  $107$ – $110^\circ$  and all bond distances in the range  $2.16$ – $2.18$  Å.

The tribromide ion in  $\text{PBr}_7$  differs, however, from that found in  $[\text{N}(\text{CH}_3)_3\text{H}^+]_2\text{Br}^-\text{Br}_3^-$  in that the two bond lengths are significantly different and the angle between the bonds is more nearly  $180^\circ$ . The  $\text{Br}(4)$ – $\text{Br}(6)$  distance of  $2.39$  Å is only slightly greater than the single-bond distance of  $2.28$  Å in  $\text{Br}_2$ , while the  $\text{Br}(4)$ – $\text{Br}(5)$  distance is lengthened to  $2.91$  Å. This is analogous to the triiodide ion, which is symmetrical in  $(\text{C}_6\text{H}_5)_4\text{AsI}_3$  (Mooney-Slater, 1959) but unsymmetrical in  $\text{NH}_4\text{I}_3$  (Mooney, 1935) and  $\text{CsI}_3$  (Tasman & Boswijk, 1955), except that the distortion is considerably greater in the tribromide case.

Havinga & Wiebenga (1959) have pointed out that in both cases of the unsymmetrical  $\text{I}_3^-$ , one iodine atom is more closely surrounded by the cation than the other two iodine atoms, whereas the surroundings of the symmetrical  $\text{I}_3^-$  are more nearly symmetrical. This is also found in the  $\text{Br}_3^-$  ion of  $\text{PBr}_7$ , where  $\text{Br}(5)$  is much closer to the bromine atoms of various  $\text{PBr}_4^+$  ions than is either  $\text{Br}(4)$  or  $\text{Br}(6)$ . In fact,  $\text{Br}(5)$  has four bromine neighbors in the  $3.1$ – $3.4$  Å range while the nearest approaches to  $\text{Br}(4)$  and  $\text{Br}(6)$  are all greater than  $3.8$  Å. Since the van der Waals radius of bromine is  $1.95$  Å, it would appear that there are significant interactions (presumably mainly ionic) between the positive  $\text{PBr}_4^+$  ion and the elongated end of the tribromide ion. It is interesting to note that each bromine atom on the

$\text{PBr}_4^+$  ion is involved in one and only one short contact with  $\text{Br}(5)$ , thus explaining why the  $\text{PBr}_4^+$  ion is so close to tetrahedral despite these interactions. These distances are shown in Table 4 and Fig. 2.

Table 4. Nearest neighbors of atoms in  $\text{Br}_3^-$

$\text{Br}(5)$ – $\text{Br}(1)$	3.14 Å
$\text{Br}(5)$ – $\text{Br}(2)$	3.26 Å
$\text{Br}(5)$ – $\text{Br}(3)$	3.36 Å
$\text{Br}(6)$ – $\text{Br}(1)$	3.88 Å
$\text{Br}(4)$ – $\text{Br}(1)$	3.89 Å

All other distances are above  $4.00$  Å.

Havinga & Wiebenga (1959) have discussed the structure of polyhalogen compounds and ions, in particular  $\text{I}_3^-$ , in terms of the LCAO molecular orbital method. The molecular orbitals were represented as linear combinations of the outer  $p$ -orbitals of the halogen atoms in the normal  $3c$ – $4e$  approximation. The Coulomb integrals were considered as a function of the electronegativity and the formal charge of the atoms, and the overlap integrals were assumed equal to zero. In the case of the unsymmetrical  $\text{I}_3^-$  ion, the Coulomb integral of the atom closest to the cations was increased by  $0.1\beta$  to take into account the effect of these cations. The most stable configurations and bond orders were qualitatively correct for the numerous cases to which this method was applied. It was therefore used in considering the configuration of  $\text{Br}_3^-$  in  $\text{PBr}_7$ . In this case, however, the modification for the effect of the cation was greater since the distortion in  $\text{Br}_3^-$  is greater than in  $\text{I}_3^-$ . Hence, the Coulomb integral on  $\text{Br}(5)$  was increased by  $0.3\beta$ . The resultant bond orders and atomic charges are shown in Fig. 3. The calculated bond orders agree

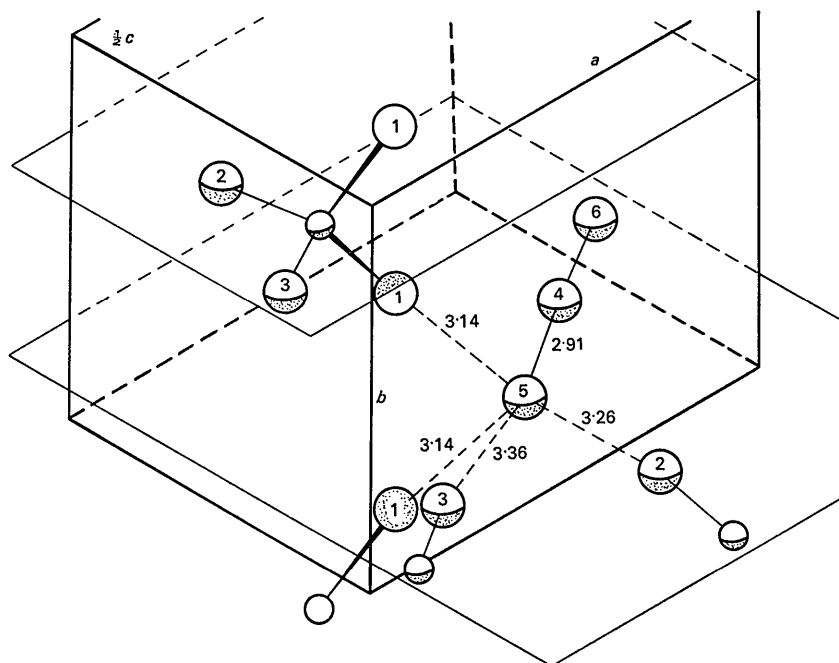


Fig. 2. Nearest neighbors of  $\text{Br}(5)$ .

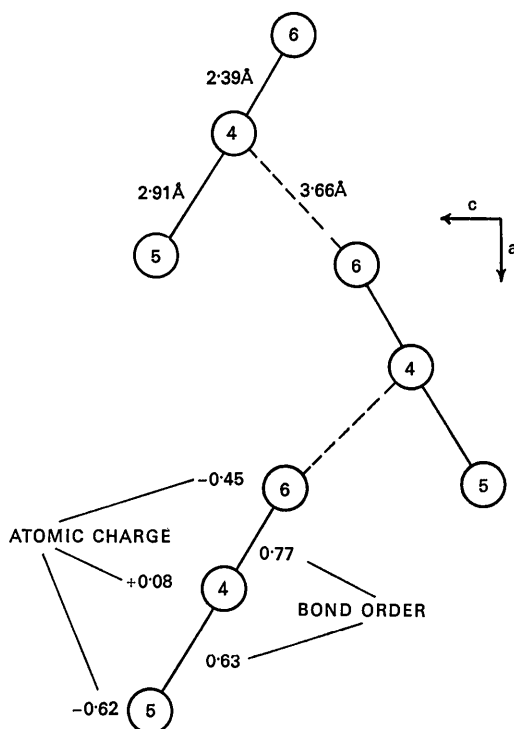


Fig. 3. Bond distances and bond orders in  $\text{Br}_3^-$ .

qualitatively with the observed bond lengths. In addition, the calculated atomic charges agree qualitatively with the preliminary results of a nuclear quadrupole resonance study of  $\text{PBr}_7$ . These results will be published when completed. The positive charge on  $\text{Br}(4)$  might explain the slight bending of the  $\text{Br}_3^-$  ion. As Fig. 3

shows, the tribromide ions form a zigzag chain through the crystal. It appears that electrostatic interaction between the positively charged  $\text{Br}(4)$  and the negatively charged  $\text{Br}(6)$  slightly bends the  $\text{Br}_3^-$  ion. The fact that the needle axis of the crystal is along the direction of the tribromide ion chain gives support to this proposed interaction between the tribromide ions.

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## Die Kristallstruktur von $\text{Sn}_2\text{S}_3$

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(Eingegangen am 11. Oktober 1966)

The existence of a new sulphide of tin,  $\text{Sn}_2\text{S}_3$ , besides the well known compounds  $\text{SnS}_2$  and  $\text{SnS}$  was confirmed, and its crystal structure determined, by a three-dimensional single-crystal X-ray diffraction analysis. The space group is  $Pnam$  with  $a = 8.864$ ,  $b = 14.020$ ,  $c = 3.747$  Å, and four formula units  $\text{Sn}_2\text{S}_3$  in the unit cell. All atoms lie in mirror planes with  $z = \frac{1}{2}, \frac{3}{2}$  [special position 4(c)].  $\text{Sn}_2\text{S}_3$  is almost isotypic with  $\text{MCdCl}_3$  with  $M = \text{NH}_4, \text{K}, \text{Rb}$ . The crystal structure consists of infinite double rutile strings of  $\text{Sn}^{\text{IV}}\text{S}_6$  octahedra parallel to the  $c$  axis, with the  $\text{Sn}^{\text{II}}$  atoms attached laterally. The geometric relationship to the crystal structure of  $\text{SnS}_2$  is discussed.

### Einleitung

Die Beobachtung von weiteren Phasen im System Zinn/Schwefel neben den bekannten Verbindungen  $\text{SnS}_2$  und  $\text{SnS}$  wird wiederholt in der Literatur erwähnt (z.B. Albers & Schol, 1961; hier finden sich auch Hinweise

auf andere Arbeiten). Die Zusammensetzungen dieser Phasen werden mit  $\text{Sn}_2\text{S}_3$  und  $\text{Sn}_3\text{S}_4$  angegeben. Die Pulverdiagramme, soweit sie mitgeteilt sind, vermitteln kein klares Bild von der Anzahl und Identität der beschriebenen Phasen, so dass die Erforschung des Systems Zinn/Schwefel noch nicht als abgeschlossen gelten kann.